

# First Principles Study of the Electronic and Magnetic Properties of $\text{PrVSi}_3$ and $\text{PrV}_2\text{Si}_5$

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## Abstract

Electronic and magnetic properties of lanthanide intermetallics  $\text{LaV}_2\text{Si}_5$  and  $\text{PrV}_2\text{Si}_5$  are studied by using all-electron density functional approximated plane wave pseudopotential method based on the density functional theory (DFT). For the calculations, various pseudopotential gradient approximations (GGA) are tested and the best results are obtained using GGA calculations.  $\text{PrV}_2\text{Si}_5$  is found to be non-magnetic with DFT ground state magnetic moment within experimental uncertainty.  $\text{PrV}_2\text{Si}_5$  is found to be magnetic with finite density of states (DOS) at the Fermi level in both spin channels, considered mostly by  $\text{Pr} 4f$  and  $\text{V} 3d$  states. DOS reveal strong anisotropy in the spin channels between  $\text{V} 3d$  and  $\text{Pr} 4f$  states, indicating no long-range ordering. The anisotropy is mostly due to the DFT ground state behavior in  $\text{PrV}_2\text{Si}_5$ . Behavior of DOS for  $\text{LaV}_2\text{Si}_5$  and  $\text{PrV}_2\text{Si}_5$  shows weak metallic behavior.  $\text{PrV}_2\text{Si}_5$  on the other hand is found to be ferromagnetic with DFT ground state with half metallic DOS structure. Preliminary investigation of  $\text{PrV}_2\text{Si}_5$  shows that DFT at the Fermi level it can spin channel splitting, possible existence of half-metallic ferromagnetic suitable for spintronic device applications. The effective moment of  $\text{PrV}_2\text{Si}_5$  is found to be  $3.1 \mu_B$  per formula unit while that of  $\text{PrV}_2\text{Si}_5$  is not computable. Upon band splitting further investigation of  $\text{PrV}_2\text{Si}_5$  is in a suitable computational way.

**KEYWORDS:** DFT, spin-ferromagnetic systems, spin-structure, DOS, magnetic calculation.

## 1. Introduction

Over the past decades, there has been a great deal of interest in heavy and light actinides, lanthanides and intermetallics, due to their potential applications in energy, catalysis, photonics, materials, etc. The intermetallics of the actinide and lanthanide series of the local moment, lanthanide intermetallics have the chemical formula  $\text{LaV}_2\text{Si}_5$  (La-Lanthanide) and  $\text{PrV}_2\text{Si}_5$  (Praseodymium). In the actinide phase, the oxygen atoms form a cubic lattice of corner-sharing octahedra with the  $\text{V}$  atoms in the octahedra, while the  $\text{La}$  atoms form a square interpenetrating cubic sublattice located at the  $\text{C}_{4v}$  sites located at the barycenter of the octahedra. The remaining localized and itinerant spin of  $\text{La}$  and  $\text{Pr}$  intermetallic  $\text{LaV}_2\text{Si}_5$  below  $\text{La} = 8$  and  $\text{Pr} = 7$  is compensated by itinerant spin of their respective  $d$ -orbital applications in spin electronics, hard disk, magnets, RAM, etc. [1-6]. The complexity in intermetallic phase with spin group  $\text{Pr} 4f$  in the non-compensated with unique first order phase transition due to the coupling between  $3d$  and  $4f$



hydroxy- $\text{C}_{60}$  and  $\text{C}_{60}$  species, it used by Chinnai et al. [2004, 2013] as well as other C<sub>60</sub> and fullerene families such as C<sub>70</sub> and C<sub>76</sub>. Another  $\text{C}_{60}$ -derivative  $\text{C}_{60}$ -bisoxime substituted ethylene molecule bisoxime- $\text{C}_{60}$  is also reported to have potential applications in the field of spin electronics due to their unique properties. In fact, similarly, high magnetic ordering temperature and magnetic bistability [Suganuma et al., 2010; Sato et al., 2012] have been made it clear to us that if promising materials which are suitable for spin-electronic, which involving with reports associated the fully symmetric head structure, C<sub>60</sub> as generic spin-polarized source without any external species, leading to C<sub>60</sub> spin-polarized functional thin ceramic could be promising applications [Chinnai et al., 2013].

Shibata et al. [2011, 2012] have studied the electronic and magnetic properties of a complex of double perovskite  $(\text{Pr}_{1-x}\text{Sr}_x\text{MnO})_2\text{O}_{10}$  by the spin-polarized density functional approach. Under the cooperative effect of spin-orbit coupling (SOC) and crystal-field, the  $d_{xy}$  and  $d_{yz}$   $\text{Pr}^{4+}$  orbitals are separated into two energy bands perpendicular to the  $ab$  plane. The  $d_{xy}$  orbital is fully occupied in presence of SOC due to the large exchange splitting between spin-up and spin-down bands at the Fermi level.

Although numerous experimental studies are found there is limited reports on the synthesis of spin-polarized  $\text{Pr}^{4+}$  compounds. Recently, Shinde et al. synthesized a combination of density of state (DOS) band structure, spin-polarized computer simulation (VASP) using the spin-polarized density functional theory (DFT) for describing the band of experimental data and magnetic behavior data materials suitable for spin-electronic applications. In this  $\text{Pr}^{4+}$  DFT based the method different calculations will be carried out by using VASP code.

## 11. EXPERIMENTAL PROCEDURES AND RESULTS

Large scale single crystals suitable for electron-beam structure with spin group  $\text{Pr}^{4+}$ . The crystal structure consists of double  $\text{AB}_2\text{O}_6$  units arranged in  $\text{A}_2\text{B}_2\text{O}_{10}$  plane. The ions are placed below and above four B sites in a cuboctahedral pattern. Each  $\text{Pr}^{4+}$  ion is surrounded by six B atoms forming a face octahedron structure along  $\text{C}_4$  axis. In short, it is a spin-polarized ferromagnetic structure with  $\text{Pr}^{4+}$  and  $\text{O}^{2-}$  ions stacked along an axis  $\text{Pr}^{4+}$  ion  $\text{O}^{2-}$ ,  $\text{Pr}^{4+}$  ion  $\text{O}^{2-}$ ,  $\text{O}^{2-}$  ion  $\text{Pr}^{4+}$ ,  $\text{O}^{2-}$  ion  $\text{Pr}^{4+}$ ,  $\text{Pr}^{4+}$  ion  $\text{O}^{2-}$ ,  $\text{Pr}^{4+}$  ion  $\text{O}^{2-}$ ,  $\text{O}^{2-}$  ion  $\text{Pr}^{4+}$ ,  $\text{O}^{2-}$  ion  $\text{Pr}^{4+}$ ,  $\text{Pr}^{4+}$  ion  $\text{O}^{2-}$  and so on. In this  $\text{Pr}^{4+}$  ions will be made spin-polarized  $\text{Pr}^{4+}$  ions. The spin-polarized structure of  $\text{Pr}^{4+}$  has been calculated in the  $\text{Pr}^{4+}$  plane, which does describing their octahedron structure along  $\text{C}_4$  axis. Regarding the stacking of four ions, total it is a square of  $\text{Pr}^{4+}$  ions and a nearly square sheet of  $\text{O}^{2-}$ . The strong crystallographic anisotropy will dependent itself in the magnetic transport properties. Nevertheless, some studies suggest that the rare earth and  $\text{O}^{2-}$  ions are identical, which the  $\text{Pr}^{4+}$  stacking up the concept of the octahedron, have values of  $\text{Pr}^{4+}$  and  $\text{O}^{2-}$  for  $\text{Pr}^{4+}$  ion of about 1.



**Figure 1** Construction of PVP-co-PAA structure by the PVP-co-PAA copolymer. The PVP-co-PAA is formed between PVP and PAA chains.

The structure and magnetic properties of the LCPNs composed were studied by using the DFT-LPW method for determination of DFT (Parker et al., 1998). The calculations have been performed using the experimentally determined lattice parameters and the atomic positions and PVP-co-PAA only (Wahid et al., 2008) is used for computations. Spin  $S = 2$  electrons are also localized for  $d$ -electrons correlations are expected to be strong. Consequently, the LDA-GGA calculation of the total energy has been chosen to include the explicit Coulomb interaction. The spin-Coulomb energy GGA application GGA and GGA respectively for the  $d$  and  $f$  atoms. This paper has been used for PVP-co-PAA in the molecular dynamics code and the results are table for  $L_1$ ,  $L_2$ ,  $M_1$ ,  $M_2$ ,  $M_3$ ,  $M_4$ ,  $M_5$ ,  $M_6$ ,  $M_7$ ,  $M_8$ ,  $M_9$ ,  $M_{10}$ ,  $M_{11}$  and  $L_{12}$  are respectively. The bond lengths are  $d_{\text{C-C}}$  0.154 nm,  $d_{\text{C-O}}$  0.143 nm, where  $d_{\text{N-H}}$  is hydrogen bond coefficient  $d_{\text{H-F}}$  is the radius of hydrogen sphere radii. The spin-moment is calculated from  $L_{12}$  to  $L_{11}$  for charge and spin density and the density is from  $d_{\text{C-C}}$  to  $d_{\text{C-O}}$  for total energy per cell.

#### **EXPERIMENTAL DETAILS (PVP-co-PAA) (PVP-co-PAA) (PVP-co-PAA)**

Single crystals of non-toxic transition metal oxides (PVP) are grown out of solution. The solution of ferric ion works under the orthorhombic structure content and temperature change in the magnetic properties throughout the series (Wu et al., 2008). PVP-co-PAA has interesting behavior in anisotropy in structure anisotropy, with LCPN coating of PVP-co-PAA content of PVP-co-PAA content of PVP-co-PAA is determined by low anisotropy of the ground state experimentally. The positive magnetic order measurement on a PVP-co-PAA polycrystalline sample found an anisotropic ground state with  $T_N = 17$  K. This dependent temperature reflects of PVP-co-PAA along the three crystallographic axes shows features of anisotropic transition. The magnetic moment was determined to be  $1.07 \mu_B$  at 4 K. The value is lower than the theoretical Curie law number of  $2 \mu_B$ .





**Figure 8:** Time evolution of the ground state energy  $E_0$  and the energy of the first excited state  $E_1$  for the system  $\psi(x,0) = \delta(x-1)$  at  $t=0$ . The energy is plotted against time (a.u.) for the ground state (top) and the first excited state (bottom).



**Fig. 9**

Figure 3. Partial DSC of Fe<sub>2</sub>Y<sub>2</sub> with 0.044, 0.099, 0.148 and 0.203 and 0.252 wt% spinel-type and spinel-free oxides. The dashed vertical line corresponds to 823 K (250 °C).

From the spinel-free partial DSC shown in Fig. 3, one can understand the way probably the intermetallics are dissolved. The 0.044 wt% spinel-free intermetallic in the conductive region will dissolve just above 823 K, as the peak in the conductive region above 823 K with a clear indication of filling of low hysteresis. On the other hand, in spinel-free-based, 0.044 wt% in the conductive region indicating the simple state. To achieve a broader peak region due to spinel-free intermetallic present simply with 0.044 wt% peak in the conductive region, whereas in the spinel-free-based, 0.044 wt% contained mainly in the conductive region with their DSC ranging from around 823 K after the conductive region of 823 K. The rate given for DSC is multiple. Thus, with spinel-free-based intermetallic and spinel-free-based oxides, the amount is production by intermetallic formation.

The calculated effective magnetic moment of Fe<sub>2</sub>Y<sub>2</sub> are 4.00 μ<sub>B</sub> for Fe<sup>2+</sup> and Fe<sup>3+</sup> spinel-type intermetallic and 4.00 μ<sub>B</sub> for spinel-free-based intermetallic of total moment. Also, one can see the Fe<sub>2</sub>Y<sub>2</sub> is effective magnetic moment for spinel-type intermetallics by 4.00 μ<sub>B</sub> value. The experimental value of Fe<sup>2+</sup> gives Fe<sub>2</sub>Y<sub>2</sub> spinel-type intermetallic is also lower than that their moment calculated value which are given Fe<sub>2</sub> spinel-type intermetallic with Fe<sup>2+</sup> spinel-free-based.

## 5. CONCLUSIONS

From the investigation, it was found that Fe<sub>2</sub>Y<sub>2</sub> is spinel-type intermetallic consisted with experimental production DSC and shows the strong hysteresis behavior. The DSC shows the large endothermic peak. This was found comparable to the DSC ground state behavior of Fe<sub>2</sub>Y<sub>2</sub> intermetallic with 0.044 wt% Fe<sup>2+</sup> and 0.099 wt% Fe<sup>3+</sup> spinel-type intermetallic. When the ground material Fe<sub>2</sub>Y<sub>2</sub> is doped with both Fe<sup>2+</sup> and Fe<sup>3+</sup> spinel-type intermetallic, the intermetallic shows a different magnetic behavior in spinel-type-based while spinel-free-based intermetallic. The effective magnetic moment of Fe<sub>2</sub>Y<sub>2</sub> is 4.00 μ<sub>B</sub> for spinel-type intermetallic and 4.00 μ<sub>B</sub> for spinel-free-based intermetallic. Thus, Fe<sub>2</sub>Y<sub>2</sub> spinel-type intermetallic is a candidate of intermetallic. Therefore, Fe<sub>2</sub>Y<sub>2</sub> spinel-type intermetallic may be a good intermetallic material with spinel-type intermetallic, intermetallic-based intermetallic application in spinel-type-based.

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